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Title: Modeling Dislocation Dynamics Near Sound Speeds in Cubic Crystals

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Modeling Dislocation Dynamics Near Sound Speeds in Cubic Crystals



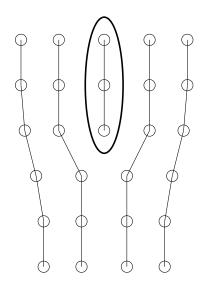
Kevin Gordon Kleiner, Daniel Blaschke (XCP-5), and Saryu Fensin (MST-8)

July 22, 2019



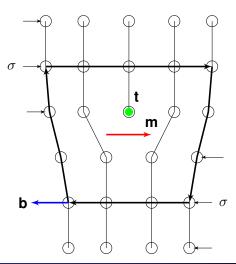
Dislocations as Defects

- Real crystals have many defects that break periodicity
- Curvi-linear defects generated and moved with plastic deformation
- Edge dislocation extra half-plane between existing planes



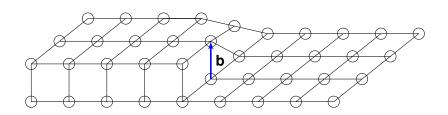
Features of Dislocations

- Circuit discontinuity not in perfect crystals - Burger's vector **b**
- Shear stress σ causes motion **m** in plane spanned by **b** and dislocation line t
- Dislocation glide due to breaking and forming bonds near the core



Screw Dislocations

- Winding of atomic planes about an axis
- Vertical jump not in perfect crystals **b** || **t** point along screw axis¹
- No single slip plane crystal geometry dependent
- **b** determined by direction of nearest neighboring atoms



¹D. Hull and D. J. Bacon, "Defects in Crystals", in: *Introduction to Dislocations*, 5th ed. Elsevier Ltd., 2011, 1-19.

High Velocity-High Stress Regime

- Little is known about dislocation glide in these conditions
- Possible speed limit by crystal sound waves
- Can dislocations be accelerated past sound speeds without losing stability?
- Material deformation rates strongly related to²
 - Mobile dislocation speeds
 - Density of mobile dislocations
 - Time derivative of that density
- Multi-physics simulation codes build in this high-level data and relevant defect speed limits

²P. Rodriguez, *B. Mater. Sci.* **19**, 6 (1996).

Molecular Dynamics (MD) Simulations

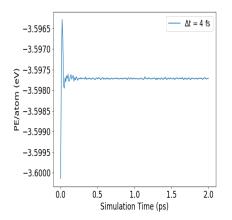
- Individual atoms move around through force evaluations³
- Effective forces based on system properties
- Energetically favorable dynamics at femtosecond resolution
- E.g., edge dislocation in face-centered cubic copper splits into partials
- Plastic glide moves both partials simultaneously⁴

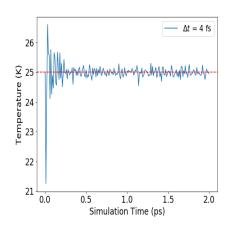
³S. Plimpton, *J. Comp. Phys.* **117**, 1-19 (1995).

⁴H. Tsuauki, P. S. Branicio, and J. P. Rino, *Appl. Phys. Lett.* **92**, 191909 (2008).

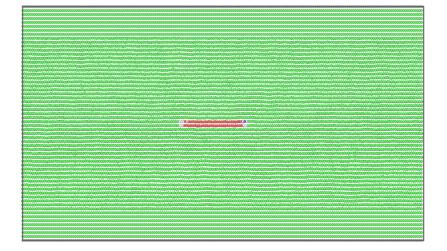
Copper Initial Equilibration for T = 25 K

- For target temperature, randomly assign atom velocities
- Stabilize configuration accordingly thermal expansion

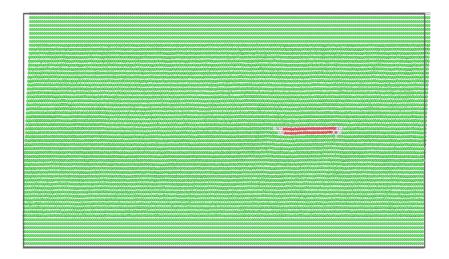




Edge Pair with σ_{yz} = 0.2 GPa at t = 0 ps

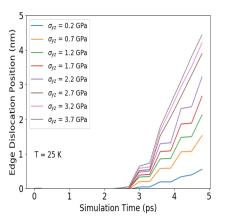


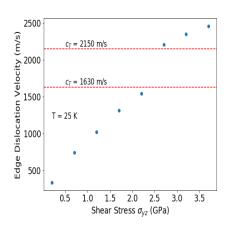
Edge Pair with σ_{yz} = 0.2 GPa at t = 4.2 ps



Edge Pair with σ_{VZ} = 0.2 GPa at t = 7.8 ps

Trajectory Dependence on Shear Stress





- Slopes calculated over last 1.8 ps
- Not steady state velocities smaller than expected in copper

Stress-Strain Relations

Glide due to plastic deformation, but an elastic model can capture relevant physics

$$\sigma_{ij} = \sum_{k} \sum_{j} C_{ijkl} \epsilon_{kl}$$

Infinitesimal strain ϵ as a gradient of continuum displacement field u

$$\epsilon_{ij} = \frac{1}{2} \left(\frac{\partial u_i}{\partial x_i} + \frac{\partial u_j}{\partial x_i} \right)$$

• Screw: $\mathbf{u} = (0, 0, u_z(x, y))$ and edge: $\mathbf{u} = (u_x(x, y), u_y(x, y), 0)$ with infinite dislocations lines in z

Introducing Anisotropy

 Cubic symmetry ⇒ three independent elastic constants $\{c_{12}, c_{44}, c_{11}\}$ for coordinates aligned with crystal axes construct stiffness tensor⁵

$$C_{ijkl} = c_{12}\delta_{ij}\delta_{kl} + c_{44}(\delta_{ik}\delta_{jl} + \delta_{il}\delta_{jk}) - (2c_{44} + c_{12} - c_{11})\sum_{\alpha}\delta_{i\alpha}\delta_{j\alpha}\delta_{k\alpha}\delta_{l\alpha}$$

 But the simplified components of the dislocation u don't align with $\{c_{12}, c_{44}, c_{11}\}$ - rotate the tensor of constants into the slip plane

$$C_{i'j'k'l'} = \sum_{i} \sum_{k} \sum_{k} \sum_{l} R_{i'i} R_{j'j} R_{k'k} R_{l'l} C_{ijkl}$$

⁵D. N. Blaschke and B. A. Szajewski, *Philos. Mag.* **98**, 26 (2018).

Continuum Dislocation Equations of Motion

$$\rho \frac{\partial^2 u_i}{\partial t^2} = \sum_j \frac{\partial \sigma_{ij}}{\partial x_j}$$

- Continuum approximation breaks down at the core (x, y) = (0, 0)
- Initial condition:
 - Dislocation stationary at t = 0
 - Dislocation accelerating for t > 0
- Boundary Conditions:
 - $-\mathbf{u}(r_0, \theta_0, z_0, t) = \mathbf{u}(r_0, \theta_0 + 2\pi, z_0, t) + \mathbf{b}$
 - Total stress field balances at dislocation core
- See if strain energy density $(1/2)\text{Tr}(\epsilon \cdot \sigma)$ remains finite around the core when accelerating to high speeds

Screw Equation of Motion

Simplify the tricky coupled system to:

$$\rho \frac{\partial^2 u_z}{\partial t^2} = \frac{\partial \sigma_{xz}}{\partial x} + \frac{\partial \sigma_{yz}}{\partial y} + \frac{\partial \sigma_{zz}}{\partial z}$$
$$\Leftrightarrow A \frac{\partial^2 u_z}{\partial x^2} + B \frac{\partial^2 u_z}{\partial x \partial y} + C \frac{\partial^2 u_z}{\partial y^2} - \rho \frac{\partial^2 u_z}{\partial t^2} = 0$$

- Constants A, B, and C depend on $\{c_{12}, c_{44}, c_{11}\}$ and rotations into the relevant crystal slip plane
- Also impose an acceleration (external stress) as:

$$x=x_0+\frac{1}{2}a_0t^2$$

Choice of Initial Condition

- Assume the isotropic static solution is sufficiently close
- $A = C = c_{44}$ and B = 0

$$c_{44}\frac{\partial^2 u_z}{\partial x^2} + c_{44}\frac{\partial^2 u_z}{\partial y^2} = 0$$

The unique solution that satisfies the Burger's circuit condition:⁶

$$u_z(x,y) = \frac{b}{2\pi} \tan^{-1} \left(\frac{y}{x}\right)$$

- Valid outside the dislocation core (x, y) = (0, 0)
- Evolve this solution for t > 0

⁶J. Weertman, "High Velocity Dislocations", in: In Response of Metals to High Velocity Deformation, Ed. P.G. Shewmon and V.F. Zackay. Interscience Publishers, New York, 1961, 205-247.

Numerical Solving Scheme: Finite Difference

- Solve for the screw u_7 over time while accelerating
- x − y space replaced by a finite sized discrete mesh
- $u_z(x, y, t) \equiv u_{i,i}^k$
- $u_z(x + \Delta x, y, t) \equiv u_{i+1, i}^k$

$$\frac{\partial^2 u_z}{\partial x^2} = \frac{u_{i+1,j}^k + u_{i-1,j}^k - 2u_{i,j}^k}{\Delta x^2} + \mathcal{O}(\Delta x^2)$$
$$n\Delta x = \frac{1}{2}a_0 t_{k+1}^2 - \frac{1}{2}a_0 t_k^2$$

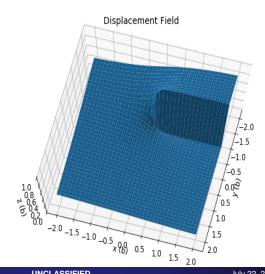
Handling Initial and Boundary Conditions

$$u_{i,j}^{k+1} = \frac{\Delta t^2}{\rho} \left(\frac{A}{\Delta x^2} (u_{i+1,j}^k + u_{i-1,j}^k - 2u_{i,j}^k) + \frac{B}{4\Delta x \Delta y} (u_{i+1,j+1}^k - u_{i+1,j-1}^k - u_{i-1,j+1}^k + u_{i-1,j-1}^k) + \frac{C}{\Delta y^2} (u_{i,j+1}^k + u_{i,j-1}^k - 2u_{i,j}^k) - \frac{\rho}{\Delta t^2} (u_{i,j}^{k-1} - 2u_{i,j}^k) \right)$$
(1)

- When k = 0, the screw is static: $u_{ii}^{-1} = u_{ii}^{0}$
- On boundaries, $u_{i,i}^{k+1} = (b/2\pi) \tan^{-1}(y_i/x_i)$ where y_i accelerates with time

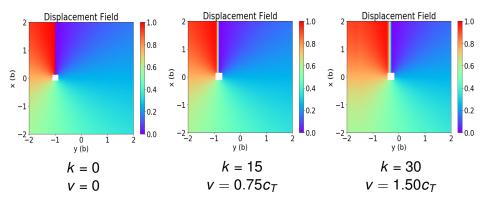
Continuum Screw Dislocation in Aluminum

- Initial form for defect in aluminum: $(b/2\pi) \tan^{-1}(y/x)$
- Models an infinitely winding screw up and down

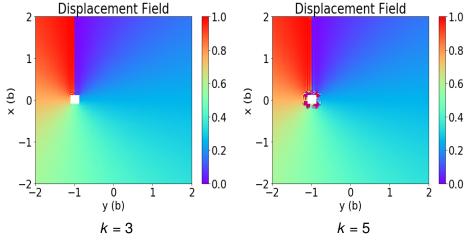


Aluminum Screw Dislocation Acceleration

- Aluminum has b = 0.286 nm and slowest $c_T = 2929 \ m/s$
- $\Delta t = 2.93 \text{ fs}$



Aluminum Screw Dislocation Solver with $a_0 = 0$



Challenge to compute second derivatives near cutout

Conclusions

- MD edge dislocation may outpace transverse sound in copper
 - Wait until steady state for more accuracy
- Continuum equation of motion solving incomplete
 - Numerical instabilities near dislocation core
 - Integrating external acceleration with the solver
 - Start with an anisotropic initial condition
- Possible collaboration opportunity for solving expertise
 - E.g., adaptive mesh refinement to treat the core differently
 - Alternatives schemes finite elements
- Once successful, different crystal structures/dislocations
- Direct calculations of deformation rates for data in hydro-codes